# Joint U.S.-Russia Conference on Advances in Materials Science August 31 – September 3, 2009 Prague, Czech Republic

# **Evolving Metallurgical Behaviors in Plutonium from Self-Irradiation**



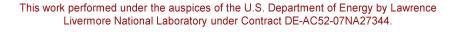
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LLNL-PRES-415264



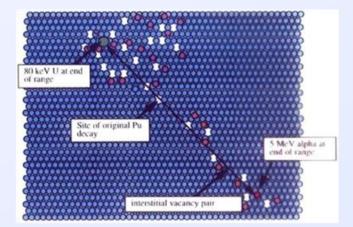




# We need to assess the impact of aging on metallurgical properties of plutonium



- Assessment of aging effects in plutonium is needed in developing predictive capabilities to describe the properties of plutonium
  - Alpha decay and the associated recoil of a uranium atom produce damage in the plutonium lattice that accumulate over time
  - We must understand how this damage affects the properties of plutonium alloys as a function of their age
- This work measures the density and mechanical property changes observed from both naturally and accelerated aged plutonium alloys
  - Dilatometry, immersion density, quasi-static tensile and compression





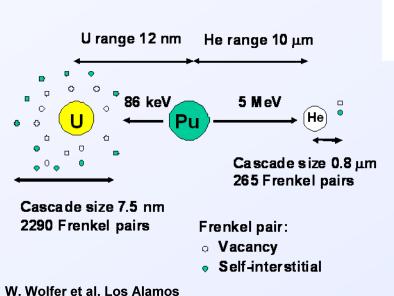




The alpha-decay in plutonium creates lattice damage, radiogenic helium and

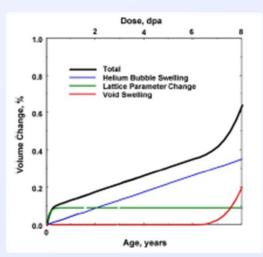
actinide daughters

 The cumulative rate of helium ingrowth is about 41 atomic parts per year



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A. Schwartz et al., Phil. Mag., v85, p479 (2005)



W. Wolfer et al., J. Alloys Compds., v444-445, p72 (2007)

Helium in-growth is generally known to be the dominant effect during extended aging



Science, v26, p274 (2000)

# We used natural and accelerated aged alloys to measure aging effects on density and mechanical properties



### **Accelerated Aged Alloy Preparation**

- We cannot afford to wait for long periods to measure aging effects in plutonium
- Accelerated the aging process by doping with <sup>238</sup>Pu
  - Damage process and defect types are identical to reference materials
- Pu alloys are enriched with approximately 7.5 wt% <sup>238</sup>Pu
  - Acceleration rate based on the difference in the activity of the spiked alloy to that of the reference alloy – Initial rate is roughly 18 fold increase
  - This rate will decrease as the material ages due primarily to decreasing concentration of <sup>238</sup>Pu in the spiked alloy

### **Density Measurements**

- Dilatometry
- Immersion density
- Aged natural alloys (PA),
   Reference alloys (RA), and
   Accelerated aged alloys (AA)

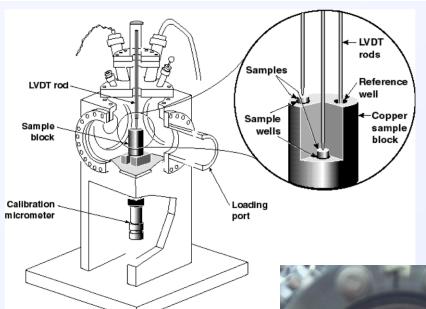
### **Mechanical Tests**

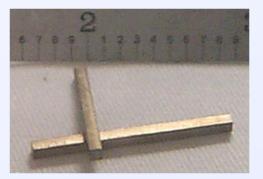
- Tensile at RT (quasi-static)
- Compression at RT (quasi-static)
- Aged natural alloys (PA), Reference alloys (RA), and Accelerated aged alloys (AA)



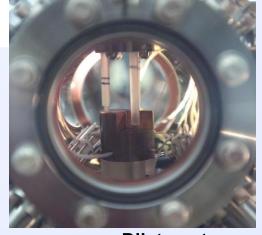
# Dilatometry System is Designed for Maximum Sensitivity and Stability







Dilatometry samples



Dilatometry system

Design based on Linear Variable Displacement Transducer (LVDT)

Two different lengths (2 and 3 cm) are used to differentiate between surface oxidation and volumetric swelling

Third well contains zero-dur as reference material

Stability over 120 days is better than

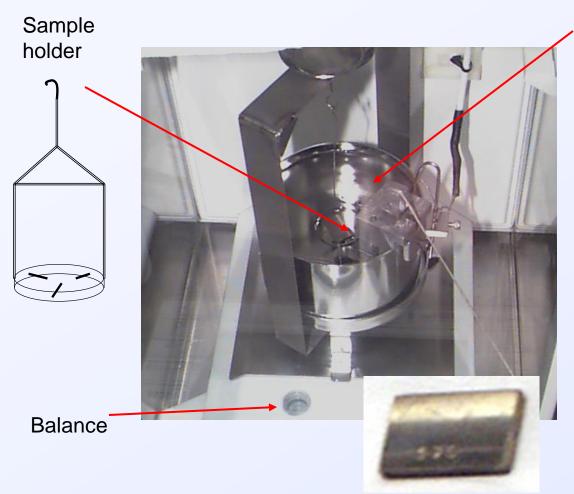
 $\pm$  0.10  $\mu m$  for the LVDTs

± 0.03°C for the Cu block



## Immersion Density Technique is Used to Measure Density of both Reference and Spiked Pu Alloys





FC-43 Inert fluorochemical as the displacement media

1.9423 g/cc, -0.0021 g/cc per °C

Prior to use, the system is calibrated using NIST glass (SRM 1827A)

A correction needs to be applied to the measured density to compensate for the heat generated by <sup>238</sup>Pu

Test equipment is based on immersion density equipment design by H. A. Bowman, et al., *J. of Res. Nat. Bur. Stand.*, <u>71C</u> (3), 179-198 (1967)



# Tensile Test Equipment and procedural methodology to meet QA compliance

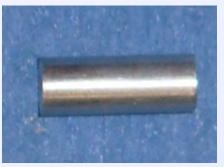


- INSTRON Model 4444 Single Column Testing System. Cap. 2kN (200kg, 450 lb)
- Series IX<sup>®</sup>Software for Windows to run tests, analyze results and store data
- Practice runs upon a comprehensive operating procedure using surrogate samples AI(1100)



### **Tensile part**

Gauge diameter 0.060 ± 0.001 inches Gauge length 0.316 ± 0.001 inches



#### **Compression part**

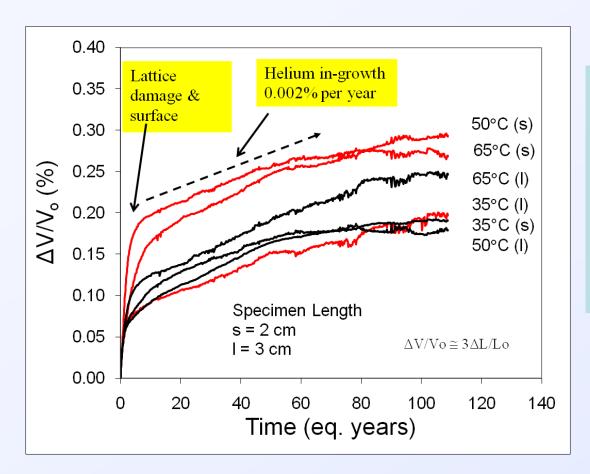
Diameter  $0.100 \pm 0.001$  inches Length  $0.300 \pm 0.001$  inches





# Dilatometry on Spiked Pu alloys with ~0.6 wt. % Ga shows volume swelling with age





- Sample temperatures maintained at 35, 50 & 65C under helium atmosphere
- Small fluctuations in data caused by external environments affecting data acquisition systems
  - Outside temperatures
  - External vibrations

Equilibrium swelling does not exceed 0.002%/year

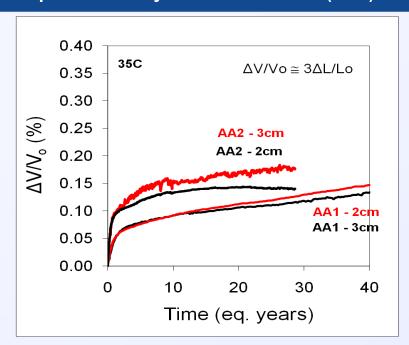
The rate of volume change decreased after ~60 eq. years



# Spiked Pu alloys with ~1 wt. % Ga at 35°C show volume swelling with age

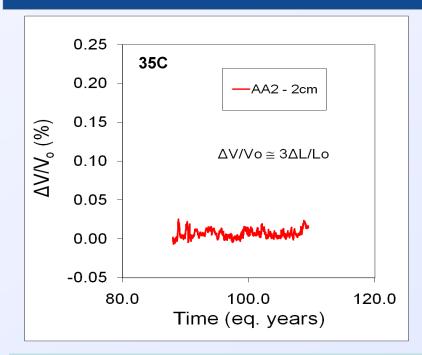


### Spiked Pu alloys with ~1wt. % Ga (AA2)



- AA2 alloys were annealed at 150°C for 30 minutes to remove accumulated lattice damage (eq. age of 86 years)
- Initial transient indicates larger change compared to AA1 alloys (~0.07% at 3 eq. years)

### Spiked Pu alloys with ~1wt. % Ga (AA2)



- AA2 alloy aged to eq. ages of 86 years does not show significant volume changes
  - Similar to aged AA1 alloys



# Dilatometry on both spiked and reference alloys shows volume swelling due to lattice damage and helium in-growth



### **Data Analysis and Findings**

- The known contributors to the inverse exponential-type and linear volume swelling are changes in lattice parameters and the build-up of helium, respectively.
  - The curves at 35°C are quite accurately represented by the combination of the exponential and linear growth

$$\Delta V/Vo = A [1-exp(-Bt)] + Ct$$

- While the curves at 50 and 65°C are influenced by possible surface oxidation during the initial stage of swelling, helium in-growth dominates the linear growth
- Helium-to-vacancy ratio can be approximated with the linear swelling rate and the helium-induced expansion relationship

$$C = (\Delta V/Vo)He/t \cong [He]/(ratio)$$

[He] = 41.1 appm per year & *t* is time

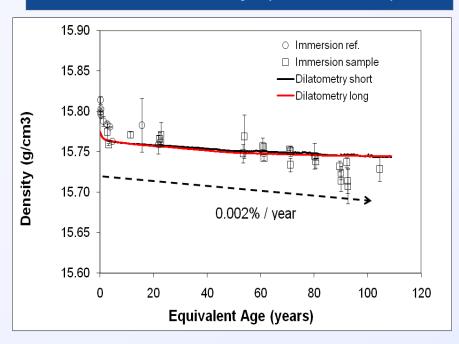
- At less than 60 eq. years, the average He/Vacancy ratio is 2.5±0.6 AA1
- At greater than 60 eq. years, the rate of volume change decreases AA1
  - Similar observation made with aged AA2



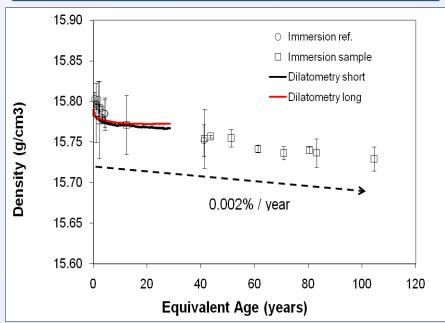
# Immersion density of spiked, reference, and naturally aged Pu alloys shows gradual decrease with age



### ~0.6 wt. % Ga alloys (AA1 and RA1)



### ~1 wt. % Ga alloys (AA2 and RA2)



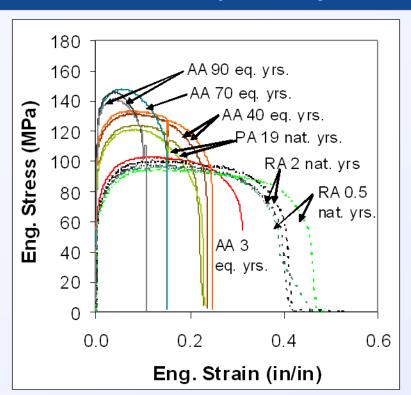
- Pu alloys show gradual decrease in immersion density with age without signs of void swelling
- Density samples are stored in a helium-flushed incubator maintained at 50°C between measurements



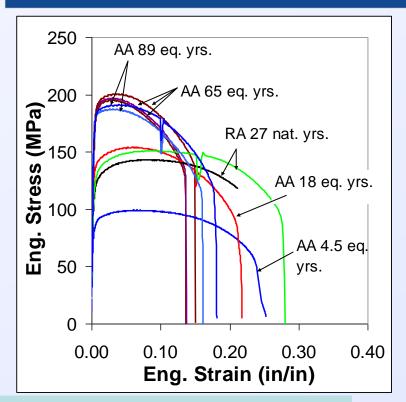
# Tensile tests on Pu alloys at ambient temperature and pressure shows increase in strength due to the aging process



### ~0.6 wt. % Ga doped Pu alloys



### ~1 wt. % Ga doped Pu alloys



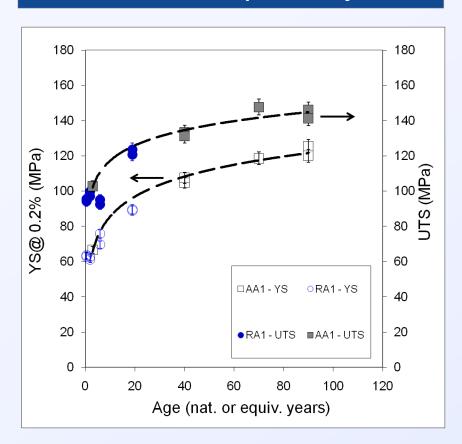
- Both alloys show increase in the yield strength (YS) and the ultimate tensile strength (UTS) with age.
- Possible saturation of YS and UTS after 70 eq. years of aging
- Slight increase in Young's Modulus, from ~40 GPa to ~45 GPa
- The ductility decreases with age



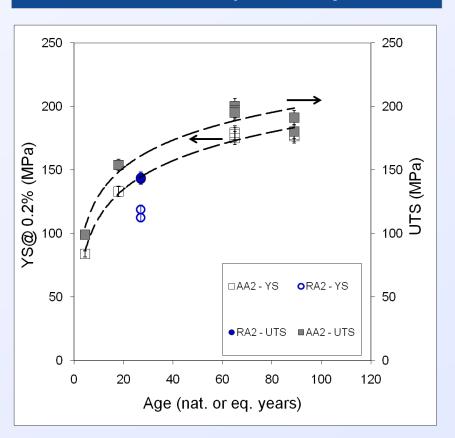
# Tensile tests show increasing yield strength and ultimate tensile strength as plutonium ages



### ~0.6 wt. % Ga doped Pu alloys



### ~1 wt. % Ga doped Pu alloys



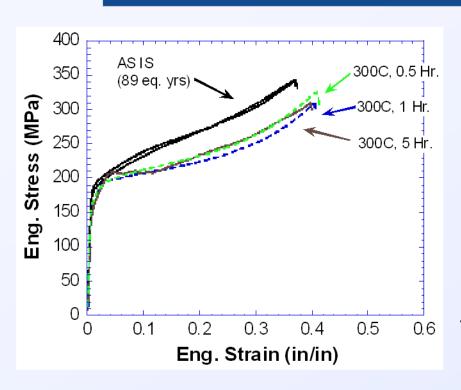
Tests are showing saturation in YS and UTS between 70 and 90 eq. yrs. of aging



# Helium in-growth contribution to static mechanical property can be studied by annealing



### Static compression tests on enriched Pu alloys (AA2)



- •The yield strength from aged (∼89 eq. yrs.) plutonium alloy is 172 MPa
  - Combined contributions from intrinsic, lattice damage, and He in-growth components
  - Intrinsic yield strength is ~80 MPa
- •The annealed yield strength is ~140 MPa
  - •Reduction of ~30 Mpa due to annealing out the accumulated lattice damage
- •The in-growth of helium contributes ∼60 MPa for this alloy aged to 89 equiv. years
  - Saturation?
- Annealing experiments can be applied to improve our understanding of aging effects to metallurgical properties of plutonium



# Metallurgical properties of spiked and naturally aged alloys evolve by aging process



- Dilatometry and immersion density show decrease in densities of <sup>238</sup>Pu spiked alloys (AA), reference (RA), and aged stockpile (PA) alloys from the aging process
  - Average He/vacant site ratio is 2.5 to 60 equivalent years, maybe changing at time greater than 60 eq. yrs.
- Void swelling is not yet observed
- The change in density calculated from the dilatometry corresponds well to the immersion density of spiked and naturally aged Pu alloys
- Static tensile and compression tests show increase in strength and decrease in ductility with age
  - Both lattice damage and helium in-growth contributes to changes in mechanical properties
    - Possible saturation effects after ~60 eq. years of aging
  - Contribution of helium in-growth to mechanical strength can be studied by annealing experiments



# Plan to continue testing both spiked and naturally aged alloys to assess the aging effects in plutonium alloys



- We need further work to determine the cause for change in ΔV/V<sub>o</sub> after aging 60 equivalent years
  - Helium bubble size, extended defects (more than simple Frenkel pairs?), transuranic in-growth, etc
- There appears to be minimal change in YS and UTS after 60 eq. yrs. of aging
  - Possible saturation effect?
- Additional annealing tests on aged samples to improve aging effects in Pu alloys
- Additional efforts in metallography and microscopy started to support density and mechanical tests

